

CORROSION BEHAVIOUR ON AA2014/35vol. %Al₂O_{3P} AND AA2009/20vol. %SiC_w COMPOSITE AFTER AGEING TREATMENT

ZAMRI BIN YUSOFF (0530410050)

A thesis submitted

In fulfilment of the requirements for the degree of

Masters of Science

School of Material Engineering UNIVERSITI MALAYSIA PERLIS

2007

ACKNOWLEDGEMENTS

Throughout my programme of study at University of Malaysia Perlis, a large number of people have given their helps and advice, either directly or indirectly. I wish to extent my sincere thanks to all those who made this programme of study possible.

In particular, I must express my sincere gratitude and appreciation to Prof. Madya Dr. Shamsul Bahrin Jamaluddin and Mr Khairel Rafezi Ahmad, my thesis supervisor and co supervisor respectively, for their scholarly guidance, encouragement and supervision in the preparation of this thesis. Without their assistance and support, this research could never had been completed.

Acknowledgement is also extended to the University of Malaysia Perlis for financial support and for the use of facilities of laboratories during the course of my study.

It is a pleasure to acknowledge the assistance given to me by the staff of the Library of University Malaysia Perlis and Polytechnic Tuanku Syed Sirajuddin Perlis, for allowing me to use all the facilities in their libraries. Thanks are also due to all the academic staff of the School of Material Engineering, Universiti Malaysia Perlis in particular to Mr. Murizam, Mr. Sobri, Mr. Nazri, Mr. Nasir, Mr. Idrus, Mr. Ku Hazrin and the technicians for their help throughout this research. Furthermore, I am also thankful to the office staffs for their helps in the administration affairs.

I wish to record my deep gratitude to a number of friends, far too many to name, who assistance and encouragement will not be forgotten, some who edited, some who printed the draft of the thesis, and others who simply in friendship inquiring about my progress.

Deep gratitude is extended to Mr. Rashdan, Mr. Hasnizam in the preparation of this thesis; they were responsible for reading the drafts. Special thanks go to my friends Mr. Rashidi and Mrs. Faridah who helped me printed the thesis.

Finally, for their prayers, patience, sacrifice, support and unfailing love, I wish to convey my sincerest appreciation to my wife, Zaimah, and my son, Ahmad Luqman as well as my daughters, Siti Khadijah, Siti Maryam, Siti Sarah and Siti Hajar.

Zamri Yusoff

Chisternis protection

Chisternis protection

CONTENTS

Topic		Pa	age
Acknowledger	nents		i
Contents			.iii
List of Tables.			vii
List of Figures			.viii
Abstract			xiv
Abstrak			.xv
CHAPTER 1	INTRO	ODUCTION	1
	1.1	The aim of research.	3
	1.2	Methodology of research	6
CHAPTER 2	REVI	DDUCTION. The aim of research. Methodology of research. EW OF LITERATURE	
	2.1	Introduction to aluminium and its alloys	7
		2.15 Metallurgical Characteristics	. 10
Chi	, xer	2.1.2 Typical Applications for several common aluminium alloys	13
© Ku.	2.2	Introduction to composites.	14
		2.2.1 Particle-reinforced composites	15
		2.2.2 Fiber-reinforced composites	15
		2.2.3. Metal matrix composites	16
	2.3	Introduction to aluminium matrix composites	17

	2.4.	Process	sing of aluminium matrix composites	21
		2.4.1	Mixing/Vortex	21
		2.4.2	Powder metallurgy(P/M)	.22
		2.4.3	Infiltration	24
		2.4.4	Rheocasting	25
		2.4.5	In-situ production	26
	2.5	Review	v of properties of aluminium matrix composites	27
	2.6.	Introdu	action to Precipitation Hardening	29
	2.7.	Introd	action to corrosion	32
		2.7.1	The Corrosion process	33
		2.7.2.	Common corrosive attack on aluminium alloys	39
			2.5.2.1 Pitting corrosion	39
		, or	2.7.2.2 Crevice corrosion	40
	xem.	2.7.3.	Corrosion behaviour of aluminium matrix composites	
1015			2.7.3.1 Pitting corrosion	44
0			2.7.3.2 Corrosion studies on AMCs	45
			n trial transmission designation	
CHAPTER 3	MAT	ERIAL	S AND METHODS	
	3.1	Mater	ials	48
		3.1.1.	Matrix material	49
		3.1.2.	Reinforcement	. 50

	3.1.3. Method of fabrication	51
	3.1.4. Specimen preparation for microstructure	
	characterisation	53
	3.2. Optical microscope	55
	3.3. Scanning electron microscope and EDX	56
	3.4. Age hardening treatment	56
	3.4. Age hardening treatment. 3.5. Microhardness measurements. 3.6. Resistivity measurements.	57
	3.6. Resistivity measurements	57
	3.7. Corrosion test	58
	3.7.1 Procedure of immersion test	58
	3.7.2. Procedure of salt spray test	60
	3.8. X-ray diffraction (XRD)	. 63
	, otect	
	.59	
CHAPTER 4	RESULTS AND DISCUSSION	
4.1.	Microstructural observations on as received composites	. 64
4.2.	Microstructural and EDX observations on	
0	as received composites	. 72
4.3.	Hardness and resistivity changes during artificial ageing	. 76
	4.3.1. Age hardening at 165°C	. 77
	4.3.2. Age hardening at 205°C	. 80
4.4.	Microstructure and EDX analyses on composites	
	after artificial ageing	85

4.5.	Gravin	netric results after immersion test	90
	4.5.1	Microstructure evaluation on specimen	
		after immersion test	102
4.6.	Gravin	netric results after salt spray test	107
	4.6.1.	Morphology and corrosion products on specimen	
		after salt spray test	119
	4.6.2.	XRD characterisation after salt spray	. 123
4.7.	Prima	ry phases during the fabrication process and	
	second	dary phases during aging	126
4.8.	Effect	of primary and secondary phases to corrosion attack	126
4.9.	Influe	ence of reinforcement.	127
4.10.	Influ	ence of fabrication route	127
4.11.	Influ	ence of voids at the particles/matrix interface	128
4.12.	The in	influence of reaction product at Al ₂ O _{3p} /AA2014 interface	128
4.13.	The it	nfluence of reaction product at SiC _w /AA2009 interface	129
4.14.	The in	nfluence of precipitates	130
405.	The in	nfluence of elements in matrix alloy	130
0			
CHAPTER 5	CON	CLUSION	. 132
BIBLIOGRA	АРНҮ		. 134
APPENDICI	E S		. 139
Appendice A	- Raw	Data	
Appendice B	B – List	of presented papers	

LIST OF TABLES

Table 2.1: Aluminium alloy designation system in the United States	8
Table 2.2: Typical application for several common aluminium alloys	13
Table 2.3: Shows the electrochemical series in various elements	35
Table 2.4: Shows the galvanic series in sea water	36
Table 3.1: Composition limits of AA2014 and AA2009	
Table 3.1: Composition limits of AA2014 and AA2009 aluminium matrix alloys. Table 3.2: Information of specimens. Table 3.3: Label of the specimens. Table 3.4: Metallography preparation method.	50
Table 3.2: Information of specimens	52
Table 3.3: Label of the specimens.	54
Table 3.4: Metallography preparation method	55
Table 4.1: List of the tested specimen composite AA2014/15vol. %Al ₂ O _{3p} ageing at 165°C and 205°C	76
Table 4.2: List of the tested specimen composite AA2009/20vol. %SiCw ageing at 165°C and 205°C	. 76
Table 4.3: List of the immersion tested specimen composite AA2014/15vol. %Al ₂ O _{3p} ageing at 165 ⁰ C	. 90
Table 4.4: List of the immersion tested specimen composite AA2009/20vol. %SiC _w ageing at 165°C	. 90
Table 4.5: List of the salt spray tested specimens	107
Table 4.6: Kinetic laws of the specimens exposed to salt spray test	. 118

LIST OF FIGURES

Figure 2.1: Useful cast and wrought aluminium alloys	9
Figure 2.2: FeAl ₃ inclusions in annealed 1100 aluminium (350x Mag.)	11
Figure 2.3: Mg ₂ Si precipitates in annealed 5457 aluminium alloy(74x Mag.)	11
Figure 2.4: Portion of the Aluminium-Magnesium phase diagram	11
Figure 2.5: Portion of the Aluminium- Copper phase diagram.	11
OX.	19
Figure 2.7: The shape of silicon carbide whiskers (SiC _w)	20
Figure 2.8: The mixing/vortex method of Al-based composites	22
Figure 2.9: The steps of the powder metal urgical processing route	23
Figure 2.10: The schematic temperature-versus-time for precipitation hardening	.30
Figure 2.11: The effect of time and temperature on precipitation process	31
Figure 2.12: The Corrosion Process	33
Figure 2.13: The surface of one component	34
Figure 2.14 The imperfect coating copper on steel	37
Figure 2.15: The corrosion of steel under a droplet of water	37
Figure 2.16: Mechanism of the crevice corrosion	40
Figure 2.17: Mechanism of pitting corrosion	41
Figure 3.1: Schematic diagram of AA2014/15vol. %Al ₂ O _{3p} as-received bar from the Duralcan, USA	48
Figure 3.2: Schematic diagram of AA2009/20vol. %SiC _w as-received bar from the ACMC, USA	49
Figure 3.3: Specimen preparation for composite AA2014/15vol. %Al ₂ O _{3p}	53

Figure 3.4: Specimen preparation for composite AA2009/20vol. %SiCw	53
Figure 3.5: Specimen hanger for immersion tests	. 59
Figure 3.6: Salt spray tester model 7004-M	. 60
Figure 3.7: The cold mounted specimens were located at the platform of salt spray test chamber	61
Figure 3.8: Interior of test chamber consist of spray tower and spray adjustor	61
Figure 4.1: The microstructure of composite AA2014/15vol. %A2O _{3p} in centre transverse section perpendicular to extrusion direction (ST direction)	65
Figure 4.2: The microstructure of composite AA2609/20vol. %SiC _w (500x magnification) in centre transverse section perpendicular to extrusion direction (ST direction)	66
Figure 4.3: The microstructure of composite AA2014/15vol. %Al ₂ O _{3p} in centre extruded surface parallel to extrusion direction (L direction).	67
Figure 4.4: The microstructure of composite AA2009/20vol. %SiCw in centre extraded surface parallel to extrusion	67
Figure 4.5: The microstructure of composite AA2014/15vol. %Al ₂ O _{3p} in extruded edge surface parallel to extrusion axis	
Figure 4.6: The microstructure of composite AA2009/20vol. %SiC _w	69
(500x magnification) in extruded edge surface parallel to extrusion axis (LT direction)	69
Figure 4.7: The microstructure of composite AA2014/15vol. %Al ₂ O _{3p} in side extruded surface parallel to extrusion direction (L direction)	70
Figure 4.8: The microstructure of composite AA2009/20vol. %SiC _w in side extruded surface parallel to extrusion direction	
(L direction)	7

Figure 4.9: SEM micrographs of AA2014 reinforced with 15 vol% alumina particles for the as received sample cut parallel to the extrusion direction	73
Figure 4.10: SEM micrographs of AA2009 reinforced with 20 vol% silicon carbide whiskers for the as received sample cut parallel to the extrusion direction	73
Figure 4.11: SEM micrographs and EDX pattern of AA2014 reinforced with 15 vol% alumina particles for the as received sample cut parallel to the extrusion direction.	74
Figure 4.12: SEM micrographs and EDX pattern of AA2009 reinforced with 20 vol% silicon carbide whiskers for the as received sample cut parallel to the extrusion direction	75
Figure 4.13: Hardness and resistivity of composite A 2014/15vol. %Al ₂ O _{3p} ageing at 165 °C	78
Figure 4.14: Hardness and resistivity of composite AA2009/20vol. %SiC _w ageing at 165 °C	79
Figure 4.15: Hardness and resistivity of composite AA2014/15vol. %Al ₂ O _{3p} ageing at 205 °C	80
Figure 4.16: Hardness and resistivity of composite AA2009/20vol.%SiC _w ageing at 205°C	. 81
Figure 4.17: SEM micrograph and EDX pattern of composite A2014/15vol.%Al ₂ O _{3p} artificially aged at 165 °C	86
Figure 4.18: SEM micrograph and EDX pattern of composite AA2009/20vol. %SiC _w artificially aged at 165 °C	87
Figure 4.19: Optical micrograph of composite AA2014/15vol. %Al ₂ O _{3p} as quenched at 1000x magnification	88
Figure 4.20: Optical micrograph of composite AA2014/15vol. %Al ₂ O _{3p} over aged 12 hours at 1000x magnification	88
Figure 4.21: Optical micrograph of composite AA2009/20vol. %SiCw as quenched at 500x magnification	89

Figure 4.22:	Optical micrograph of composite AA2009/20vol. %SiCw	
	over aged 12 hours 100x magnification	89
Figure 4.23:	Mass loss versus time for unaged sample composites	
	AA2014/15vol. %Al ₂ O _{3p} and AA2009/20vol. %SiC _w	
	respectively after immersed in 3.5 wt% NaCl for 28 days	94
Figure 4.24:	Mass loss versus time for aged sample composites	
	AA2014/15vol. %Al ₂ O _{3p} and AA2009/20vol. %SiC _w	
	ageing at 165 °C about 2 hrs and then immersed to a	
	3.5 wt% NaCl for 28 days	95
Figure 4.25:	Mass loss versus time for aged sample composites	
	AA2014/15vol.%Al ₂ O _{3p} and AA2009/20vol.%SiC _w	
	ageing at 165 °C about 4 hrs and then immersed in	
	3.5 wt% NaCl for 28 days.	96
Figure 4 26	: Mass loss versus time for aged sample composites	
1 igure 4.20	AA2014/15vol.%Al ₂ O _{3p} and AA2009/20vol.%SiC _w	
	ageing at 165 °C about 8 hrs and then immersed to a	
	3.5 wt% NaCl for 28 days	97
F: 4.07	W 1	
Figure 4.27	: Mass loss versus time for aged sample composites AA2014/15vol. %Al ₂ O _{3p} and AA2009/20vol. %SiC _w	
	ageing at 165 °C about 12 hrs and then immersed in	
	3.5 wt% NaCl for 28 days	98
Figure 4.28	: Mass loss versus time for unaged sample composite	
	AA2014/15vol. %Al ₂ O _{3p} and aged specimen composite	
	AA2014/15vol. %Al ₂ O _{3p} ageing at 165 °C about 12 hrs	
	and then immersed in 3.5 wt% NaCl for 28 days	. 100
•	5	
Figure 4.29	Mass loss versus time for unaged composite	
	AA2009/20vol. %SiC _w and aged specimen composite	
0	AA2009/20vol. %SiC _w ageing at 165 °C about 12 hrs	
	and then immersed in 3.5 wt% NaCl for 28 days	. 101
Figure 4.30): Pitting corrosion attack after 28 days immersion in	
	the matrix through the interface Al ₂ O ₃ p /matrix in the	
	composite AA2014/15vol. %Al ₂ O _{3p} in shallow wide pit shape	102
Figure 4.3	1: Pitting corrosion attack after 28 days immersion in the	
	matrix through the interface Al ₂ O ₃ p /matrix in the composite	1000
	AA2014/15vol. %Al ₂ O _{3p} in vertical grain attack pit shape	103

Figure 4.32:	Pitting corrosion attack after 28 days immersion in the matrix through the interface SiCw/matrix in the composite AA2009/20vol. %SiC _w in shallow deep pit shape 103
First II	Pitting corrosion attack after 28 days immersion in the matrix through the interface SiC/matrix in the composite A A 2009/20vol %SiC in shallow deep pit shape
	AA2009/20vol. %SiC _w in shallow deep pit shape
Figure 4.34:	Common pit shape of trough pits group and sideway pits group 105
Figure 4.35:	Holes due to intensive matrix corrosion in the composite AA2014/15vol. %Al ₂ O _{3p}
Figure 4.36:	Holes due to intensive matrix corrosion in the composite AA2009/20vol. %SiC _w
Figure 4.37	The graph of weight gain (g/m²) versus exposure time after the composite AA2014/15vol. % Al2O _{3p} and composite AA2009/20vol. %SiC _w with 180 grit surface roughness exposed to 3.5 wt% salt spray
Figure 4.38	: The graph of weight gain (g/m²) versus exposure time after the composite AA2014/15vol. %Al ₂ O _{3p} and composite AA2009/20vol. %SiC _w with 180 grit surface roughness exposed to 5 wt% salt spray
Figure 4.39	e: The graph of weight gain (g/m²) versus exposure time after the composite AA2014/15vol. %Al ₂ O _{3p} with 600 grit surface roughness exposed to 3.5 wt% and 5 wt% salt spray
	2: The graph of weight gain (g/m²) versus exposure time after the composite AA2009/20vol. %SiC _w with 180 grit surface roughness exposed to 3.5 wt% and 5 wt% salt spray
Figure 4.41	: The graph of weight gain (g/m²) versus exposure time after the composite AA2014/15vol. %Al ₂ O _{3p} with difference surface roughness exposed to 3.5 wt% salt spray
Figure 4.42	2: The graph of weight gain (g/m²) versus exposure time after the composite AA2009/20vol. %SiC _w with difference surface roughness exposed to 5 wt% salt spray
Figure 4.43	3: SEM micrographs of composite (a) AA2014/15vol. %Al ₂ O _{3p} exposed to 3.5 wt% NaCl (b) AA2014/15vol. %Al ₂ O _{3p} exposed to 5.0 wt% NaCl for 20 hrs. The surface roughness of
	the specimen is P180 grit120

Figure 4.44: SEM micrographs of composite (a) AA2009/20vol. %SiC _w exposed to 3.5 wt% NaCl and (b) AA2009/20vol. %SiC _w exposed to 5.0 wt%NaCl for 20 hours. The surface roughness of the specimen is P180 grit
Figure 4.45: SEM micrographs of composite AA2014/15vol. %Al ₂ O _{3p} after exposed to 3.5 wt% NaCl for 20 hrs. The surface roughness of specimens are (a) P180 grit (b) P400 grit and (c) P600 grit1122
Figure 4.46: Shows the qualitative diffraction pattern from composite AA2014/15vol. %Al ₂ O _{3p} after 20 hrs exposure (surface roughness P180 – 5 wt% NaCl)
Figure 4.47: Shows the qualitative diffraction pattern from composite AA2009/20vol. %SiC _w after 20 hrs exposure (surface roughness P180 – 5 wt% NaCl)
AA2009/20vol. %SiCw after 20 hrs exposure (surface roughness P180 – 5 wt% NaCl)
in the same of the
Sproze
· Sixell
O This

ABSTRACT

The purpose of this study is to compare the corrosion behaviour between AA2014/15vol. %Al2O3p and AA2009/20vol. %SiCw composites after ageing treatment. The former composite was cast composite supplied by Duralcan USA. The later composite was a powder metallurgy composites supplied by Advanced Composite Materials Corporation USA. The corrosion behaviour was investigated by means of immersion test and salt spray test. Before immersion test, the composites were examined for ageing response at 165°C (0 - 12hours). Microstructure evaluation has been characterized using optical microscope, SEM and EDX. It was observed that the distribution of SiC whiskers in the matrix AA2009 is more homogeneous compared to the distribution of Al₂O₃ particles in the matrix AA2014. The secondary phases such as Al₂Cu and Al₂CuMg were identified in AA2014/15vol. %Al₂O_{3p} whereas Al₂Cu was identified in AA2009/20vol. %SiCw composites after ageing. Localized corrosion is more intense in A2014/15vol. %Al2O3p compared with AA2009/20vol. %SiCw. The pit shape for composite of AA2014/15vol. %Al2O3p is shallow wide pit and differs to composite AA2009/20vol. %SiCw which is shallow deep pit. Salt spray test has been used to assess the corrosion behaviour of the surface of as received composites. The corrosion product has been analysis using XRD after salt spray test. The qualitative diffraction pattern from both composites after 20 hours exposure in salt spray test shows the formation of the corrosion products. It was observed that three corrosion products were identified such as Mg (OH) 2, Al2O3, MnO2 in composite AA2014/15vol. %Al2O3p and no corrosion product was identified for composite AA2009/20vol. %SiCw.

ABSTRAK

antara komposit Kajian bertuiuan membandingkan kelakuan kakisan AA2014/15vol.%Al2O3p dengan komposit AA2009/20vol.%SiCw. Komposit pertama dibekalkan oleh Duralcan, USA yang difabrikasikan melalui kaedah tuangan adukan. Komposit kedua dibekalkan oleh Advanced Composite Materials Corporation, USA yang difabrikasikan melalui kaedah metalugi serbuk. Kelakuan kakisan dikaji menggunakan ujian rendaman dan ujian semburan garam. Kesan penuaan ke atas kedua-dua komposit pada suhu 165°C (0 hingga 12jam) telah dikaji sebelum ujian rendaman dijalankan. Penilaian mikrostruktur dibuat menggunakan mikroskop optikal, mikroskop imbasan electron dan analisis EDX Adalah diperhatikan bahawa taburan SiC_w dalam matrik AA2009 lebih homogen berbanding taburan Al₂O_{3p} dalam matrik AA2014. Keputusan SEM menunjukkan kebanyakan partikel dan wisker bertaburan secara hampir selari dengan arah penyempritan. Pembentukan mendakan Al2Cu dan Al2CuMg didapati dalam AA2014/15vol.%Al2O3p manakala Al2Cu didapati dalam AA2009/20vol.%Sicw selepas penuaan. Kakisan setempat lebih teruk dalam AA2014/15vol.%Al2O3p berbanding AA2009/20vol.%SiCw. Bentuk bopeng pada komposit AA2014/15vol.%Al2O3p adalah bopeng meluas sebaliknya komposit adalah bopeng mendalam. Ujian semburan garam telah AA2009/20vol.%SiCw digunakan untuk menilai kelakuan kakisan dan hasil karat pada permukaan komposit. Hasil kakisan di analisa menggunakan XRD. Keputusan analisa XRD ke atas keduadua komposit selepas 20 jam pendedahan dalam semburan garam menunjukkan adanya hasil kakisan seperti Mg(OH)2, Al2O3, MnO2 dalam komposit AA2014/15vol.%Al2O3p dan tiada hasil kakisan dalam komposit AA2009/20vol.%SiCw.

CHAPTER 1

INTRODUCTION

The future of the fundamental and hi-tech industries depends on the development and improvement of engineering materials. For example, many aerospace applications required materials with high thermal conductivity to reduce component operating temperature and moderate density to reduce overall system weight. Since the 1960s, metals have been strengthened and stiffened by the incorporation of reinforcement. In the 1970s and 1980s, a considerable amount of research was expended in improving the mechanical properties of aluminium alloys (Budinski and Budinski, 2005).

Metal matrix composites (MMCs) are widely used in structures where a high strength to weight ratio is prime concern, such as in the transportation industry. MMCs, particularly with ceramic particles or whiskers reinforcing aluminium- based alloys, have been extensively investigated since 1980s (Borrego, 1998). Aluminium matrix composites (AMCs) are currently being developed for engineering components because of high stiffness, good elevated temperature properties and good fatigue resistance (Trowsdale, 1996). Research on the aluminium matrix composites is still at the development stage, but the outlook is very promising. In the next 20 years, these materials are expected to overtake conventional materials as base alloys. In recent years, the aerospace, military and automotive industries have been promising the technological

development of composite materials to achieve good mechanical strength/density and stiffness/density ratios (Pardo et al, 2005).

To date, much of the researches are focused on the mechanical properties and correlations between fabricating route, microstructure and properties of AMCs. If AMCs are to become incorporated into current engineering systems, their corrosion behaviour must be determined and investigated before employing them to service environments. Therefore, corrosion behaviour studies on aluminium matrix composites in marine environments and the influence of the ageing, matrix composition, type of reinforcement and the chloride concentration should be conducted properly.

The limitation of AMC application is restricted due to the effect of the reinforcement on the corrosion resistance. The addition of the reinforcing materials to aluminium may influence the susceptibility to degradation. Reinforcement materials should possess significantly higher specific stiffness and strength, as well as high melting and decomposing temperatures compared to the base alloy. The ease of processing and part fabrication is key factors in determining the selection of the reinforcement form and type. The major reinforcements used in aluminium based MMCs are boron, graphite, silicon carbide and alumina. Majority of the current investigations appear to be focusing on SiC and Al₂O₃ reinforced Al MMCs.

Discontinuous reinforcement appears to influence the corrosion behaviour of MMCs not only by acting as nucleation site for primary and secondary phase precipitation during fabrication and heat treatment, but also hindering the formation of a coherent nonporous oxide layer (Lucas, 1992). Both factors are believed to be largely responsible for the corrosion site in the matrix and Al/reinforcement interface. The exact nature of the electrochemical events at corrosion site on MMCs is still under debates (Lucas, 1992). Therefore, the present study is carried out in order to understand the events leading to the corrosion site of aluminium matrix composite before and after ageing using immersion test as well as optical microscope, SEM and EDX. Whereas, salt spray and X-ray methods used in order to studied the corrosion products original

The aim of research. 1.1

Several previous studies on properties of aluminium matrix composite have been reported focusing on either comparative study between monolithic alloy and reinforced alloy in terms of influence of similar type of reinforcement or comparative study between different reinforced alloys in terms of similar type of reinforcement. Furthermore, several studies have focused on the properties of aluminium matrix composites from the same fabrication route such as casting and powder metallurgy. Some researchers studied the influence of heat treatment based on single aluminium matrix composite without comparison. Corrosion resistance of MMCs is also a subject of study to be discussed in order to compare their corrosion resistance towards corrosive environment and some studies have reported based on the effect of heat treatment to the corrosion behaviour.

Research works on the mechanical properties and corrosion behaviour of AA2009/20vol. %SiC_w have been reported by Chan and Tong (2000) and Yue (2000) respectively. Whereas, research works on mechanical properties and corrosion behaviour of AA2014/15vol. %Al₂O_{3p} have been reported by Molina et al (2004) and Lucas & Clarke (1992).

To date, there is no study carried out focusing on the corrosion behaviour between two different aluminum matrix composites in term of different type of reinforcement and fabrication route. Comparative study between different composites is required in order to develop information for all metal matrix composites. This information is very important for the user to select the best material for engineering applications with different properties required including its corrosion resistance. This thesis deals with the study of the corrosion behaviour of AA2014/15vol. %Al₂O_{3p} and AA2009/20vol. %SiC_w after artificial against under sodium chloride solution. AA2014/15vol. %Al₂O_{3p} was cast composite supplied by Duralcan USA. The particle size is 9 – 13 μm. AA2009/20vol. %SiC_w was a powder metallurgy composites supplied by Advanced Composite Materials Corporation (ACMC), USA. The dimension of beta-SiC whiskers reinforcement is 0.45 – 0.65 μm in diameter and 5 – 80 μm in initial length. Both of the composites were extruded by hot extrusion process into rectangular bar.

In the present research work, the corrosion behaviour of AA2014/15vol. %Al₂O_{3p} and AA2009/20vol. %SiC_w were investigated by immersion test and salt spray test in different chloride concentration. The objective was to study the corrosion rate and to clarify the influence of ageing, reinforcement, matrix composition and chloride concentration on the corrosion behaviour.

Studies on the acceleration of ageing kinetics in both aluminima matrix composites conducted by heat treatment process and monitored by scanning electron microscope (SEM), resistivity and hardness measurement. This thesis focuses on comparative study on pitting corrosion behaviour between AA2014 Svol. %Al₂O_{3p} and AA2009/20vol. %SiC_w in terms of:

- 1. Influence of different matrix element composition
- 2. Influence of different reinforcement
- 3. Influence of different fabrication method
- 4. Influence of artificial aging
- 5. Influence of chloride concentration and surface roughness

The corrosion studies include pitting, corrosion behaviour, corrosion product and pit shape.

1.2 Methodology of research

This research project used AA2014 (Al-Cu) matrix reinforced with 15vol. % of Al₂O₃ particles and AA2009 (Al-Cu) matrix reinforced with 20vol.% SiC whiskers. Methods of corrosion test used are immersion test and salt spray test. The immersion test was performed in 3.5%NaCl solution for both specimens at different time exposure at room temperature. Before corrosion test, the specimens were aging treatment under certain conditions. The salt spray test was performed in 3.5%NaCl and 5.0%NaCl solution for both specimens ant different time exposure at 50°C temperature.

Microstructure characterisation was examined using optical microscope, scanning electron microscope (SEM) and energy dispersive x-ray (EDX). Hardness measurement was conducted using vickers microfardness. The specimens were prepared according to the ASTM E.3-01 (2001) Standard Guide for preparation of Metallographic Specimens for microstructure characterisation. Scanning Electron Microscopy was used to study the microstructure of MMCs. SEM micrograph offers considerable advantages over optical micrograph. In this technique, information about morphology and reinforcement/matrix interface obtained. The energy dispersive X-ray spectroscopy was used to analyse the chemical composition and phase formation on the surfaces. X-ray diffractor meter (XRD) was used to analyze the microstructure and micro-chemical comprehensively. The XRD offers solution encompassing wide-ranging analysis requirements, from routine qualitative to quantitative analysis.

CHAPTER 2

REVIEW OF LITERATURE

2.1 Introduction to aluminium and its alloys

Aluminium and its alloys are characterised by a relatively low density (2.7 g/cm³ as compared to iron 7.9 g/cm³), high electrical and thermal conductivities, and good resistance of corrosion in some common environments. These alloys are easily formed by virtue of high ductility. Since aluminium has an FC6 crystal structure, its ductility is retained even at very low temperatures. The limitation of aluminium is its low melting temperature (660°C), which restricts the maximum temperature at which it can be used (Callister, 2000)

Aluminium alloys can be divided into two major groups: wrought and casting alloys, depending on their method of fabrication. Wrought alloys, which are shaped by plastic deformation, have compositions and microstructures significantly different from casting alloys, reflecting the different requirements of the manufacturing process. Within each major group we can divide the alloys into two subgroups: heat-treatable and non heat-treatable alloys (Askeland & Phule, 2004).

The most commonly used alloy designation system in the United States is that of the Aluminium Association. The designations for groups of wrought alloys and groups of cast alloys are shown in Table 2.1.

Table 2.1: Aluminium alloy designation system in the U.S.A. (Budinski & Budinski, 2005).

Wrought Alloy Designation	Major Alloying Elements
1xxx	Commercially pure aluminium (99% min)
2xxx	Copper (major alloying element)
3xxx	Manganese
4xxx	Silicon
5xxx	Magnesium
6xxx	Magnesium and silicon ×
7xxx	Zinc
8xxx	Other elements
9xxx	Unused series
	08
Cast Alloy Designation	Major Alloxing Elements
1-99(old system)	Aluminium + silicon
1xx.x	99.5 min. aluminium
2xx.x	Copper
3xx.x	Silicon + copper or magnesium
4xx.x	Silicon
5xx.x	Magnesium
6xx.x	Unused series
7xx.x	Zinc
5xx.x 6xx.x 7xx.x 8xx.x 9xx.x	Tin
9xx.x	Other element
6,	

For wrought alloys, it is based on four digits corresponding to the principal alloying element as shown in Table 2.1. The first digit indicates the major alloying element. The second digit in this system indicates chemistry modification or impurity. In 1xxx series third and fourth digit indicate % of Al concentration or purity of the alloy. In other series third and forth digit identified the particular alloy. Cast alloys have been identified by a four-digit identification number with last digit separated by decimal. A letter prefix is occasionally used to signify alloy or impurity limits. The first digit indicates the alloy group. The second and third digits identify an alloy within a group,